



Bi-induced p-type conductivity in nominally undoped Ga(AsBi)

G. Pettinari, A. Patanè, A. Polimeni, M. Capizzi, Xianfeng Lu, and T. Tiedje

Citation: Applied Physics Letters **100**, 092109 (2012); doi: 10.1063/1.3690901 View online: http://dx.doi.org/10.1063/1.3690901 View Table of Contents: http://scitation.aip.org/content/aip/journal/apl/100/9?ver=pdfcov Published by the AIP Publishing

Articles you may be interested in

Increased p-type conductivity in GaN x Sb1- x , experimental and theoretical aspects J. Appl. Phys. **118**, 085708 (2015); 10.1063/1.4929751

Increased p-type conductivity through use of an indium surfactant in the growth of Mg-doped GaN Appl. Phys. Lett. **106**, 222103 (2015); 10.1063/1.4922216

Electrical and optical properties of p-type InGaN Appl. Phys. Lett. **95**, 261904 (2009); 10.1063/1.3279149

Electrical isolation of p-type GaAs layers by ion irradiation J. Appl. Phys. **91**, 6585 (2002); 10.1063/1.1469693

Hole activation from GaAs:Zn nanoclusters for p-type conduction in ZnSe Appl. Phys. Lett. **76**, 1701 (2000); 10.1063/1.126141



euse of AIP Publishing content is subject to the terms at: https://publishing.aip.org/authors/rights-and-permissions. Download to IP: 142.104.83.231 On: Wed, 20 Apr 2016

Bi-induced *p*-type conductivity in nominally undoped Ga(AsBi)

G. Pettinari,^{1,a)} A. Patanè,¹ A. Polimeni,² M. Capizzi,² Xianfeng Lu,^{3,b)} and T. Tiedje³ ¹School of Physics and Astronomy, University of Nottingham, Nottingham NG7 2RD, United Kingdom ²CNISM-Dipartimento di Fisica, Sapienza Università di Roma, P.le A. Moro 2, 00185 Roma, Italy ³Department of Electrical and Computer Engineering, University of Victoria, Victoria, British Columbia V8W 3P6, Canada

(Received 30 January 2012; accepted 12 February 2012; published online 1 March 2012)

We report *p*-type conductivity in *nominally undoped* GaAs_{1-x}Bi_x epilayers for a wide range of Bi-concentrations ($0.6\% \le x \le 10.6\%$). The counterintuitive increase of the conductivity with increasing *x* is paralleled by an increase in the density of free holes by more than three orders of magnitude in the investigated Bi-concentration range. The *p*-type conductivity results from holes thermally excited from Bi-induced acceptor levels lying at 26.8 meV above the valence band edge of GaAs_{1-x}Bi_x with concentration up to 2.4×10^{17} cm⁻³ at x = 10.6%. © 2012 American Institute of Physics. [http://dx.doi.org/10.1063/1.3690901]

The unique electronic properties and uncommon dependence on composition of the band structure in dilute bismide alloys are attracting increasing interest and are boosting several exciting lines of research in Materials Science.¹ Indeed, the large size of Bi-atoms leads to significant relativistic corrections^{2,3} and a strong perturbation of the GaAs host band structure parameters, such as the band-gap energy^{4,5} and carrier effective mass.^{6,7} These properties make dilute bismides of technological interest for several applications that span from spintronics to high-efficiency solar cells,⁸ heterojunction bipolar transistors,⁹ terahertz,¹⁰ and infrared devices.^{11–13}

Despite numerous studies on dilute bismides, the transport properties of this material system are still largely unknown and debated. Previous studies of $GaAs_{1-x}Bi_x$ (x < 2.5%) have shown that the Bi incorporation does not degrade significantly the electron transport properties of GaAs (at least for $x \le 1.4\%$).^{14,15} This finding is in line with the introduction by Bi-atoms of energy levels resonant in, or close to, the GaAs valence band,¹⁶ thus affecting mainly the hole transport. The first experimental studies of the effects of Bi on the hole mobility (μ_h) have been reported only recently by Hall measurements on *p*-type (C- or Be-doped) $GaAs_{1-x}$ Bi_x samples ($x \le 1.2\%$, see Ref. 17; and $x \le 5.5\%$, see Ref. 18) and by picosecond transient-grating technique experiments on undoped GaAs_{1-x}Bi_x (2.5% $\leq x \leq 6.3\%$, see Ref. 19). These works have revealed a general decrease of the hole mobility following Bi incorporation. On the other hand, the behavior of μ_h at high x (~10%) as well as the influence of Bi on the free-hole concentration (p) is still not clear. Kini et $al.^{17}$ reported a large decrease of p with increasing x (about one order of magnitude for $x \sim 1\%$) in Cand Be-doped $GaAs_{1-x}Bi_x$ and ascribed this decrease to the effect of Bi_{Ga} heteroantisite defects, which act as double donors and compensate partially the extrinsic p-type doping of the alloy. In contrast, Nargelas et al.¹⁹ reported *p*-type conductivity in *undoped* $GaAs_{1-x}Bi_x$ samples

(2.5% $\leq x \leq$ 6.3%), thus suggesting an acceptor behavior for Bi in GaAs, but without reporting any compositional dependence of *p* or $\mu_{\rm h}$.

In this letter, we report *p*-type conductivity in *nominally* undoped GaAs_{1-x}Bi_x epilayers for a wide range of Biconcentrations ($0.6\% \le x \le 10.6\%$). Hall effect measurements reveal a monotonic increase of *p* with increasing *x*, which is accompanied by an increase of the conductivity. These results support the existence of Bi-induced acceptor levels,^{3,19} whose density increases with *x*. In the same concentration range, the free-hole mobility exhibits an overall decrease, with values in good quantitative agreement with those in the literature.

We investigate a series of $GaAs_{1-x}Bi_x$ epilayers (x = 0.6%, 3.8%, 5.6%, 8.5%, and 10.6%; thickness t = 30-56 nm) grown by solid source molecular beam epitaxy on a semi-insulating (100) GaAs substrate. Further details of the growth conditions and sample parameters can be found elsewhere.^{5,7} The Bi-concentration was determined by combining x-ray diffraction and optical data. All samples were processed into 1-3-3-1 Hall bars of width $W = 45 \mu m$ and length $L = 1200 \mu m$ with Ti/Au metal depositions providing ohmic contacts [see inset in Fig. 2(b)]. Hall effect measurements were performed using a high impedance system with high current stability (~50 pA) in a superconductive-magnet cryostat with magnetic field (*B*) up to 14 T and temperatures (*T*) ranging from 65 K to 290 K.



FIG. 1. Compositional dependence of the longitudinal conductivity σ measured in nominally undoped GaAs_{1-x}Bi_x epilayers at different temperatures.

© 2012 American Institute of Physics

euse of AIP Publishing content is subject to the terms at: https://publishing.aip.org/authors/rights-and-permissions. Download to IP: 142.104.83.231 On: Wed, 20 Apr 201

^{a)}Electronic mail: giorgio.pettinari@nottingham.ac.uk.

^{b)}Present address: Varian Semiconductor Equipment Associates, Gloucester, Massachusetts 01930, USA.



FIG. 2. (Color online) (a) Hall voltage $V_{\rm H}$ acquired at fixed magnetic field (B = 14 T) as a function of the longitudinal current *I* for different values of the Bi-concentration. (b) $V_{\rm H}/I$ as a function of *B* for different *x* (I = 0.3 nA, 2.5 nA, 60 nA, 35 nA, and 400 nA for x = 0.6%, 3.8%, 5.6%, 8.5%, and 10.6%, respectively). Inset: optical-microscope image of a 1-3-3-1 Hall bar employed for transport measurements. The dashed lines indicate, in both panels, the slope for the linear Hall effect from free holes.

Figure 1 shows the longitudinal conductivity (σ) at B = 0 T for GaAs_{1-x}Bi_x epilayers with different *x*. The data reveal a significant increase of σ with increasing *x*, which becomes steeper with decreasing *T*. The inhomogeneity in the conductivity along the Hall bar was measured to be less than 20% in all samples. The Bi-induced increase of σ differs from the systematic reduction of conductivity for increasing alloying reported for other highly mismatched alloys, such as Ga(AsN) and In(AsN).^{20,21} We investigate further this counterintuitive behavior by Hall effect measurements.

For each sample, the Hall voltage $(V_{\rm H})$ was acquired in two different experimental conditions: either as a function of the longitudinal current (I) at fixed B (14 T) or as a function of B at fixed I. In both configurations and in all samples, $V_{\rm H}$ exhibits a linear dependence on I and B, see Fig. 2. The sign of $V_{\rm H}$ is consistent with a dominant *p*-type conductivity. The values of the free-hole concentration $p = (IB)/(eV_{\rm H}t)$ (where e is the electron charge and t is the thickness of the $GaAs_{1-x}$ Bi_x epilayers) are shown in Fig. 3(a) as a function of x at T = 250 K. Within the experimental uncertainty,²² the same values of p are obtained by measuring the Hall voltage at fixed or variable magnetic field. The measured free-hole concentration increases monotonically with increasing x, reaching values of $\sim 10^{17} \text{ cm}^{-3}$ at x = 10.6%. This indicates the existence of Bi-induced acceptors and supports previous findings by different techniques; see Refs. 3 and 19. However, such an acceptor behavior of Bi-atoms is incompatible with the presence of a large density of donor levels due to Bi_{Ga} heteroantisites suggested for *p*-type doped $GaAs_{1-x}Bi_x$



FIG. 3. (Color online) (a) Compositional dependence of the free-hole concentration *p* in GaAs_{1-x}Bi_x at T = 250 K, as obtained from the analysis of $V_{\rm H}$ acquired for fixed (dots) and variable (circles) *B*. Inset: comparison between experimental data of *p* and concentration of Bi-clusters of 1, 2, and 3 Bi-atoms as estimated for a random alloy by including next-nearest-neighbor interactions (solid lines), after Ref. 24. (b) As panel (a) for the free-hole mobility, $\mu_{\rm h}$. Data of $\mu_{\rm h}$ available in the literature are reported by stars (Refs. 17–19, at T = 300 K). The dashed line is a guide to the eye.

with $x \le 1.2\%$.¹⁷ Therefore, Bi-related donor states, if present, should become rapidly negligible with increasing *x*.

The Hall mobility of free holes, $\mu_{\rm h} = \sigma/(ep)$, is shown in Fig. 3(b) as a function of x at T = 250 K. Our results (dots and circles) agree well with those reported in the literature (stars), thus providing a general overview of the compositional dependence of $\mu_{\rm h}$ over a wide range of x.²³ Although the different sets of data show some scatter, a clear-cut decrease of $\mu_{\rm h}$ is observed for x up to ~8.5%; whilst $\mu_{\rm h}$ increases from 0.9 ± 0.1 cm²/(Vs) to 3.7 ± 0.8 cm²/(Vs) for x going from 8.5% to 10.6%. This compositional dependence of $\mu_{\rm h}$ agrees with recent magneto-optical studies on the same series of samples.⁷ These measurements highlight a strong hybridization between the Bi-related levels and the host band-states for x < 6%. In turns, this results into an increase of the carriers effective mass, paralleled here by a decrease of the free-hole mobility. On the contrary, the softening of this hybridization for x > 8% leads to a decrease of the carriers effective mass, paralleled here by an increase of the free-hole mobility.

A deeper insight into the mechanism responsible for the *p*-type conductivity in $GaAs_{1-x}Bi_x$ was obtained from a detailed temperature-dependent study of μ_h and *p* at x = 10.6%, as



FIG. 4. (Color online) (a) Temperature-dependence $(65 \text{ K} \le T \le 290 \text{ K})$ of the free-hole mobility μ_h in GaAs_{1-x}Bi_x with x = 10.6%, as obtained from the analysis of V_H acquired for fixed (dots) and variable (circles) *B*. The dashed line represents the $T^{3/2}$ -dependence for impurity scattering. (b) As panel (a) for the free-hole concentration, *p*. An analysis in terms of free holes thermally activated from a single acceptor level reproduces well the experimental data for $E_b = 26.8 \text{ meV}$ and $p_0 = 2.4 \times 10^{17} \text{ cm}^{-3}$ (see solid line). The contribution of the intrinsic carrier concentration n_i is largely negligible (see dotted line).

reported in Fig. 4. The free-hole mobility increases monotonically with increasing T, as shown in Fig. 4(a), thus indicating that the contribution of elastic collisions by alloy disorder and/or defects is significantly larger than that due to inelastic collisions with phonons, at least for x = 10.6%. The dependence of p on temperature is shown in Fig. 4(b). It is well described by the exponential law $p = p_0 \times \exp(-E_b/k_{\rm B}T)$, where $E_b = 26.8 \pm 0.5 \text{ meV}$ is a single activation energy and $p_0 = (2.4 \pm 0.1) \times 10^{17} \text{ cm}^{-3}$ is the density of acceptor levels. The value of E_b agrees with the binding energy of Bi-induced acceptor levels (~25 meV) determined by far-infrared absorption spectroscopy,³ whereas the value of p_0 points towards a significant concentration of acceptors. This gives strong evidence that the *p*-type conductivity in $GaAs_{1-x}Bi_x$ results from holes thermally excited into the valence band from Bi-induced acceptor levels. The intrinsic carrier contribution to the conductivity is more than 5 orders of magnitude smaller than the measured values and can be neglected even at room temperature, see dotted line in Fig. 4(b).

In the inset of Fig. 3(a), the compositional dependence of p is compared with that of the density of clusters composed by 1, 2, or 3 Bi-atoms, as calculated for a random distribution of impurities.²⁴ This comparison indicates that low-order Bi-clusters cannot account for the compositional evolution of the acceptor density. The fast, exponential-like increase of p with x suggests the existence of a driving force ruling the formation of the acceptor centers during the growth process that cannot be explained in a pure random framework. Theoretical investigations are now required to address the geometry and number of Bi-atoms forming these clusters or defect levels.

In conclusion, we have shown that the *p*-type conductivity of *nominally undoped* $GaAs_{1-x}Bi_x$ epilayers is enhanced by Bi atoms. This behavior differs fundamentally from that found in other highly mismatched alloys and suggests that the Bi-incorporation in GaAs leads to the formation of acceptor levels lying at $\sim 27 \text{ meV}$ above the valence band edge with concentration of up to $\sim 2 \times 10^{17} \text{ cm}^{-3}$ at x = 10.6%. The systematic increase of the free-hole concentration with increasing Bi-concentration is paralleled by a non-monotonic decrease of the free-hole mobility, the latter being explained in terms of the hybridization of the Bi-induced localized states with the extended band-states of GaAs. These findings provide evidence for an unique compositional dependence of the hole transport properties of $GaAs_{1-x}Bi_x$ that will stimulate further theoretical works on the electronic properties of dilute bismides. The Bi-induced increase of conductivity is also of technological interest in the context of current research on the device applications of dilute bismides.

This work was supported by the EU under grant agreement No. PIEF-GA-2010-272612 and by the COST Action MP0805. We are grateful to O. Makarovsky and D. Taylor for their kind assistance during the transport measurements and Hall bar processing.

- ¹T. Tiedje, E. C. Young, and A. Mascarenhas, Int. J. Nanotechnol. 5, 963 (2008).
- ²B. Fluegel, S. Francoeur, A. Mascarenhas, S. Tixier, E. C. Young, and T. Tiedje, Phys. Rev. Lett. 97, 067205 (2006).
- ³G. Pettinari, H. Engelkamp, P. C. M. Christianen, J. C. Maan, A. Polimeni, M. Capizzi, X. Lu, and T. Tiedje, Phys. Rev. B 83, 201201(R) (2011).
- ⁴J. Yoshida, T. Kita, O. Wada, and K. Oe, Jpn. J. Appl. Phys. **42**, 371 (2003).
 ⁵X. Lu, D. A. Beaton, R. B. Lewis, T. Tiedje, and Y. Zhang, Appl. Phys. Lett. **95**, 041903 (2009).
- ⁶G. Pettinari, A. Polimeni, M. Capizzi, J. H. Blokland, P. C. M. Christianen, J. C. Maan, E. C. Young, and T. Tiedje, Appl. Phys. Lett. **92**, 262105 (2008).
- ⁷G. Pettinari, A. Polimeni, J. H. Blokland, R. Trotta, P. C. M. Christianen, M. Capizzi, J. C. Maan, X. Lu, E. C. Young, and T. Tiedje, *Phys. Rev. B* 81, 235211 (2010).
- ⁸F. Dimroth, Phys. Status Solidi C 3, 373 (2006).
- ⁹P. M. Asbeck, R. J. Welty, C. W. Tu, H. P. Xin, and R. E. Welser, Semicond. Sci. Technol. **17**, 898 (2002).
- ¹⁰K. Bertulis, A. Krotkus, G. Aleksejenko, V. Pačebutas, R. Adomavičius, G. Molis, and S. Marcinkevičius, Appl. Phys. Lett. 88, 201112 (2006).
- ¹¹R. B. Lewis, D. A. Beaton, X. Lu, and T. Tiedje, J. Cryst. Growth **311**, 1872 (2009).
- ¹²W. L. Sarney, S. P. Svensson, H. Hier, D. Donetsky, D. Wang, L. Shterengas, S. Suchalkin, and G. Belenky, AIP Conf. Proc. **1416**, 59 (2011).
- ¹³Y. I. Mazur, V. G. Dorogan, M. Schmidbauer, G. G. Tarasov, S. R. Johnson, X. Lu, S.-Q. Yu, Zh. M. Wang, T. Tiedje, and G. J. Salamo, Nano-technology **22**, 375703 (2011).
- ¹⁴D. G. Cooke, F. A. Hegmann, E. C. Young, and T. Tiedje, Appl. Phys. Lett. **89**, 122103 (2006).
- ¹⁵R. N. Kini, L. Bhusal, A. J. Ptak, R. France, and A. Mascarenhas, J. Appl. Phys. **106**, 043705 (2009).
- ¹⁶M. Usman, C. A. Broderick, A. Lindsay, and E. P. O'Reilly, Phys. Rev. B 84, 245202 (2011).
- ¹⁷R. N. Kini, A. J. Ptak, B. Fluegel, R. France, R. C. Reedy, and A. Mascarenhas, Phys. Rev. B 83, 075307 (2011).
- ¹⁸D. A. Beaton, R. B. Lewis, M. Masnadi-Shirazi, and T. Tiedje, J. Appl. Phys. **108**, 083708 (2010).

- ¹⁹S. Nargelas, K. Jarašiūnas, K. Bertulis, and V. Pačebutas, Appl. Phys. Lett. 98, 082115 (2011).
- ²⁰S. Fahy, A. Lindsay, H. Ouerdane, and E. P. O'Reilly, Phys. Rev. B 74, 035203 (2006).
- ²¹A. Patanè, G. Allison, L. Eaves, N. V. Kozlova, Q. D. Zhuang, A. Krier, M. Hopkinson, and G. Hill, Appl. Phys. Lett. **93**, 252106 (2008).
- ²²The uncertainties on the values of $\mu_{\rm h}$ and *p* are calculated taking into account the conductivity inhomogeneity along the Hall bar and measure-

ments in three different points on the bar, as well as measurements on two Hall bars. $V_{\rm H}$ is determined by averaging between positive and negative values of *I* and *B*.

- ²³We estimate the difference in temperatures at which μ_h has been measured in our work (T = 250 K) and in Refs. 17–19 (T = 300 K) to produce variation on μ_h within the experimental uncertainty, therefore not affecting the comparison between different sets of data.
- ²⁴M. M. Kreitman and D. L. Barnett, J. Chem. Phys. 43, 364 (1965).